# Evaluating the Wear of Polycrystalline Diamond Compact Drill Bit Cutters using Indentation and Scratch Tests

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Abstract-Polycrystalline diamond compact (PDC) drill bits are widely used in oil and gas drilling. The wear of PDC cutters is a major problem during drilling. It leads to severe time losses which affect the overall drilling operation cost. Therefore, it is essential to evaluate the wear tendency for these cutters using predictive approaches. The present research is focused on studying the wear mechanisms of PDC cutters and the effect of their mechanical properties on the extent of wear. The volume of wear for the PDC cutters was determined experimentally using micro- and nanoscratch tests by implementing an approach based on the geometry of the removed material after micro- and nano-scratch tests. The experimental wear results were compared to the predictions from current models in the literature. Various wear models are evaluated for micro- and nano-scratch tests on both layers of the PDC samples. The study shows that the wear of the PDC cutters can be predicted from the material mechanical properties, applied load, sliding distance, and hardness of the PDC cutters. The study could be extended for the evaluation of wear intensity of PDC cutters from various manufactures without using the previous techniques of abrasion testing.

Index Terms—Bit wear, Indentation, Micro-scratch test, Nano-scratch test, Polycrystalline diamond compact.

### I. Introduction

Drilling for oil and gas is still very demanding in many areas around the world. The use of polycrystalline diamond compact (PDC) bits is widely spread nowadays due to their high performance and durability in harsh environments. Synthetic diamond cutters or inserts that are placed on the PDC bit represent the most significant part of these bits. PDC cutters have significantly increased drill bit efficiency and drilling performance to a point where PDC bits have overtaken tricone bits in many oil and gas drilling

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applications. Since the first commercial introduction of PDC bits in 1976, many challenges appeared that sometimes halted their applicability, especially in hard geological formations, where abrasive wear and impact damage prevail. Wear of drilling bits is due to macroscopic or microscopic removal or fracture of material, especially at the cutter surface.

Mouritz and Hutchings (1991) investigated the wear rates of the materials used in the teeth of the rotary drilling bits and the abrasive wear mechanisms of these materials. Richardson (1968) showed that abrasive wear rate of any ductile material depended on the ratio of its hardness (Hm) to that of the abrasive (Ha). When Ha/Hm <1, this means that the abrasive cannot scratch the material and the wear rate is extremely low. However, when Ha/Hm >1.2, the abrasive can scratch the material and cause high wear rate (Mouritz and Hutchings, 1991). Geoffroy (1999) showed that more than 50% of the energy of the drilling by PDC bits is dissipated by the wear of cutters under normal drilling conditions, assuming a steady motion and avoiding excessive heating. Fang, et al. (2001) described that the main failure modes of PDC cutters are due to frictional heat and abrasiveness of the rock as shown in Fig. 1.

Tze-Pin, et al. (1992) and Zacny (2012) postulated four primary failure modes of PDC cutters according to the type of wear mechanism: 1 - smooth wear, 2 - microchipping, 3 - gross fracturing or spalling, and 4 - delamination.

Various tests have been carried out on PDC bits and, especially, their cutters to assess their performance and applicability in diverse drilling conditions including harsh environment and high-temperature conditions. Bellin, et al. (2010a) mentioned that the majority of the previous tests are broadly destructive. Abrasion test is the well-known one, where the abrasion resistance of the cutter against abrasive rocks is evaluated. It depends mainly on the hardness of the minerals in the rock. Highly abrasive rocks like granite or hard sandstone are normally used in the abrasion test. The wear of the PDC inserts has been evaluated and reported in the literature (Bellin, et al., 2010a); nevertheless, the precise assessment of the material loss remains a challenge. Therefore, the current study is focused on this topic, that is, determining the wear volume of the diamond layer as well



as the substrate of the PDC cutters after applying micro- and nano-mechanical testing.

## II. METHODOLOGY

A series of micro- and nano-indentation tests were carried out to determine the mechanical properties of the materials forming the PDC samples. The samples are in the form of cylindrical disks with different dimensions. The images of four PDC cutters that have been used in the experiments are shown in Fig. 2. The specifications of the PDC cutters supplied from two different manufacturers are given in Table I. The upper part of the PDC cutters is made of diamond layer in the form of PDC with cobalt as a binder, whereas the lowest layer (substrate) consists of tungsten carbide embedded in cobalt as a binder. PDC specimens were finely polished to reduce the effect of surface asperities and consequently to reduce the measurement error of hardness and Young's modulus. Furthermore, well-polished surfaces will produce clearer imprints from micro- and nanoindentations and enable better measurements of the diagonals

TABLE I SPECIFICATIONS OF THE PDC SPECIMENS

| Commercial order | Cutter diameter (mm) | Cutter height (mm) | Diamond thickness (mm) |
|------------------|----------------------|--------------------|------------------------|
| M1308            | 13.44 (±0.03)        | 8.0 (±0.1)         | 2.2 (±0.2)             |
| M1313            | 13.44 (±0.03)        | 13.2 (±0.1)        | $2.2 (\pm 0.2)$        |
| K1608            | $16.00 (\pm 0.05)$   | $8.0 (\pm 0.1)$    | 2.0-2.5                |
| K1908            | 19.05 (±0.05)        | 8.0 (±0.1)         | 2.0-2.5                |

PDC: Polycrystalline diamond compact

of the obtained indents. Following the above characterization work, micro- and nano-scratch tests were carried out, and the results were compared with the predictions from the wear models in the literature.

#### III. EXPERIMENTAL WORK

#### A. Micro- and nano-indentation tests

Micro- and nano-indentation measurements of hardness and Young's modulus of many materials have been performed by numerous researchers; nevertheless, many previous attempts for indenting the diamond layer of the PDC cutter have failed. Many researchers mentioned the failure to be due to the breakage of the indenter, and especially, the Berkovich type (Sumiya and Irifune, 2004). Nevertheless, the use of Berkovich probe was successful under very low loads, that is, <3 mN (Couvy, et al., 2011). In our work, nano-indentation hardness and Young's modulus of the diamond layer are measured using a Berkovich indenter using the NanoTest machine manufactured by Micro Materials Ltd., Wrexham, UK. A typical load-penetration plot for the indentation is shown in Fig. 3. Hardness and Young's modulus of the sample were obtained using the approach reported in the literature (Couvy, et al., 2011) but at different settings.

After completing the nano-indentation tests of the diamond layer, it is crucially important to check that the Berkovich probe has not been damaged. To do this, nanoindentation tests were carried out on a ceramic sample before and after the indentation process of the diamond layer. It was observed

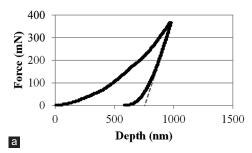


Spalling Chipping and heat Breakage checking

Fig. 1. Typical failure modes of PCD inserts for rock drill bit (Fang, et al., 2001).



Fig. 2. Oil polycrystalline diamond compact cutters manufactured by two different manufacturers.



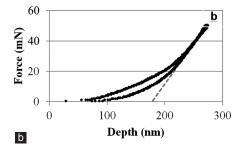


Fig. 3. (a) Applied force against displacement for tungsten carbide-cobalt at fixed load of 350 mN, (b) load versus the indentation depth for the polycrystalline diamond compact-Co at fixed load of 50 mN.

that the hardness and Young's modulus of ceramic before and after were nearly the same and this indicated that the tip of the nanoindenter was intact and not damaged.

The previous reported value of the hardness and Young's modulus of PDC was within the range 50-60 GPa and 776-925 GPa, respectively, depending on the diamond grain size and the percentage of the cobalt binder which varied from 5% to 20% (Dubrovinskaia, 2006), (Osipov, et al., 2010), and Ndlovu (2009). To prevent large fluctuation in the measurement of material properties, microindentation experiments were carried out. A series of microindentation tests were performed on the diamond and the substrate layers of the PDC samples using INSTRON 5566 equipment using Vickers indenter at various loads. Microindentations on diamond coatings require a highly polished test surface to discard the effect of the asperities that might cause undesirable results such as producing unclear indents. Fig. 4 illustrates the indents left after using Vickers probe on the substrate layer of sample K1908 after applying a load of 50 N. Table II demonstrates the average results of the hardness and Young's modulus determined from micro- and nano-indentation tests on both layers of the PDC samples. It is obvious that the standard deviation of the nanoindentation results is much higher than that of microindentation due to the aforementioned reason.

Fracture toughness was determined from the microindentation tests for the diamond and the substrate layers of the PDC inserts.

# B. Microstructural properties and the cobalt content of PDC cutters

This part of the work is carried out to study the effect of the microstructural properties of the cutters on the mechanical properties of the diamond and substrate layers. The microstructural properties have been characterized by energy dispersive X-ray (EDX) images. The particles size of the coating (diamond) and the substrate (tungsten-carbide) has been quantified by the Image J software, whereas the weight percentage of the binder (cobalt) in the PDC samples has been determined from EDX element table content and from longitudinal measurement of the binder (cobalt) across the interface between the diamond and the substrate layers using (EDX-Quantline).

Backscattered detector (BSD) images are used for displaying the best resolution of the particle distribution

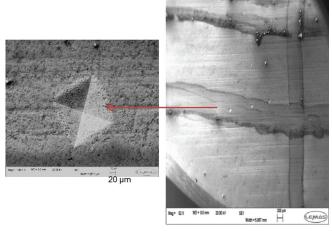


Fig. 4. Vickers imprints on the substrate of sample K190 after applying 50 N load.

compared to normal scanning electron microscope (SEM) images. EDX maps are used for determining the cobalt content in the diamond and the substrate layers of the PDC cutters.

The circular equivalent diameter of the diamond particles and substrate is determined by image J software after analyzing the BSD images. It should be noted that major drawback of this technique is that the size characteristics are based on two-dimensional (2D) image analysis and may not represent the particles third dimension.

Fig. 5 (top) shows the diamond particles (black areas) embedded in the cobalt binder (white areas) for samples M1313 (a) and K1908 (b). The particle size distribution of the diamond grains of samples M1313 and K1908 is shown in Fig. 5 (bottom).

Fig. 6 (top) illustrates the tungsten-carbide particles embedded in cobalt (blue) for (a) sample M1313 and (b) sample K1908, whereas Fig. 6 (bottom) demonstrates the tungsten-carbide grain size distribution for (c) specimen M1313 and (d) sample K1908.

The cobalt content represented as weight percentage is determined from EDX element content at various spectrum areas of the diamond and substrate layers, and then, averaged results with standard deviation are presented in Table III.

To summarize the microstructural obtained results, Table III shows the microstructural properties of two PDC samples from different manufacturers.

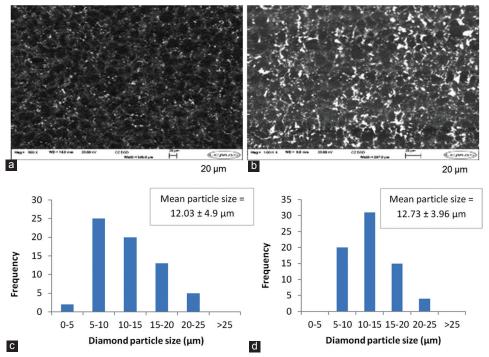


Fig. 5. (Top) Backscattered detector images. (a) sample M1313, (b) sample K1908 of the diamond layer displaying the diamond grains (black) and the cobalt (white), (bottom) diamond particle size distribution of sample M1313, and (c) sample K1908 (d).

TABLE II

MECHANICAL PROPERTIES OF PDC SAMPLES FROM MICRO- AND NANO-INDENTATION TESTS

| PDC sample                | Vickers hardness (GPa) | Yong's modulus (GPa) | Fracture toughness (MPa.m <sup>1/2</sup> ) | Layer     |
|---------------------------|------------------------|----------------------|--|-----------|
| M1313 (Nano indentation)  | 16.69±6.59             | 487.96±161.18        | N/A  | Substrate |
| M1313 (Microindentation)  | 13.072±0.066           | 516.943±2.169        | 11.908±2.152                               | Substrate |
| K1908 (Microindentation)  | 12.564±0.257           | 514.344±0.877        | 12.277±2.146                               | Substrate |
| M1313 (Nanoindentation)   | 49.36±23.80            | 807.53±181.98        | N/A  | Diamond   |
| M1313 (Microindentation)  | 50.85±0.912            | 864.534±0.986        | 9.13±1.219                                 | Diamond   |
| K1908 (Micro indentation) | 46.092±0.754           | 855.63±4.474         | 9.641±1.064                                | Diamond   |

PDC: Polycrystalline diamond compact

 ${\bf TABLE~III}$  Structural properties of PDC inserts from different manufacturers

| Microstructural properties | PDC samples  |                |  |
|----------------------------|--------------|----------------|--|
|                            | M1313        | K1908          |  |
| Diamond particle size (μm) | 12.03±4.9    | 12.73±3.96     |  |
|                            | min<4.52     | min<6.9        |  |
|                            | max>23.25    | max>23.9       |  |
| WC particle size (μn)      | 2.52±2.11    | $2.3 \pm 1.69$ |  |
|                            | min<0.38     | min<0.3        |  |
|                            | max>8.23     | max>7.73       |  |
| Co wt. % (in (diamond)     | $3.26\pm0.5$ | 4.4±0.51       |  |
| Co wt. % (in WC-Co)        | 7.37±0.39    | 8.82±0.32      |  |

PDC: Polycrystalline diamond compact

Increasing the cobalt content for a given particle size causes the fracture toughness to increase, but it usually reduces the hardness (Bellin, et al., 2010c). It is also reported that fine tungsten carbide grain size and a low level of cobalt content lead to higher hardness and good wear resistance, whereas coarse particles and high cobalt content produce low hardness and high wear rate (Mori, et al., 2003; Ndlovu,

2009). Yahiaoui, et al. (2013) stated that the grain size and the content of cobalt are the most significant parameters of PDC cutters as high wear rate is associated with large particle size. Bellin, et al. (2010b) reported that the increasing the cobalt content in the diamond layer and the substrate will increase the fracture toughness, whereas the wear resistance increases with lower cobalt content.

The particle size of the substrate layer of samples M1313 and K1908 is around 2.3-2.5  $\mu$ m, whereas the cobalt content in the substrate of sample M1313 is slightly lower than sample K1908. Accordingly, it is expected from these results that the abrasion (wear) resistance of sample M1313 is slightly greater than sample K1908, whereas sample K1908 has a better fracture toughness than sample M1313 (as can be seen from Table II).

The hardness of the diamond layer of sample M1313 found to be greater than the hardness of the sample K1908 as shown in Table II. This is mainly due to a lower level of binder content (3.26%) in sample M1313 compared to sample K1908 (4.4%), whereas the grain sizes are not that different.

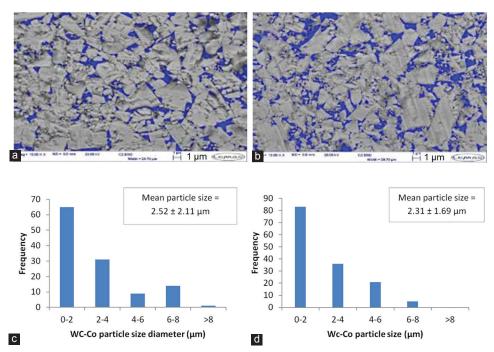


Fig. 6. (Top) BSD images (a) sample M1313, (b) sample K1908 of the substrate layer displaying the tungsten-carbide particles (gray) and the cobalt (blue), (bottom) substrate grain size distribution, (c) Specimen M1313, and (d) specimen K1908.

For the substrate layer, the hardness of the tungstencarbide layer of sample M1313 is found to be higher than sample K1908 as shown in Table II, and this is attributed to the difference in the cobalt content of the substrate for the two PDC cutters. The cobalt mass weight percentage in sample M1313 is near 7%, whereas sample K1908 has cobalt content of about 9%. The mean grain size of samples M1313 and K1908 is very similar.

# C. Microscratch testing

The work presented in this section is focused on quantifying the wear of PDC cutters and the effect of their mechanical properties on the extent of wear. The volume of wear for the PDC cutters was determined experimentally using microscratch and later from nanoscratch tests on different layers of PDC cutters, that is, the diamond layer and tungsten-carbide (substrate layer) by implementing an approach based on the geometry of the removed material after the tests. In addition, this section illustrates how the experimental wear is compared to the predicted wear based on current approaches in the literature. Various wear models are evaluated for micro- and nano-scratch tests on both layers of the PDC samples: Substrate and diamond.

The scratch caused by a sharp indenter produces abrasion wear that would be crucially important for wear mechanism and evaluation. A series of microscratch tests were carried out by applying various loads on the surface of the material and moving the probe to a sufficient sliding distance to produce a scratch. Scratch length and width are the most points of interest for evaluating the wear of the material.

The microscratch tests were performed using an INSTRON 5566 mechanical testing machine and a diamond indenter.

Vickers and spheroconical diamond indenters were used during the microscratch tests on both layers of the PDC samples. The obtained microscratches were examined by SEM, EDX, 2D Talysurf, and three-dimensional interferometer to determine the scratches width and depth as well as for analytical study.

A series of microscratch tests using a Vickers indenter were conducted on both layers of the PDC specimens under various loads. The samples were mounted on an anvil that has a manual controller to alter the sliding distance, whereas the applied load was kept constant. The substrate of the PDC sample was scratched using loads ranging 100-300 N. Measured wear from microscratch tests was compared to the predicted wear based on models in the literature.

The height of the scratch is measured by the 2D profilemeter. Careful measurements were carried out using Ultra Precision Talysurf PGI800 with a resolution up to 0.8 nm.

The volume of wear  $(V_w)$  is determined experimentally from the geometry of the removed material using the following equation:

$$V_w = \frac{w_s \cdot h_s}{2} X \tag{1}$$

Where  $w_s$  is the width of the scratch and is measured by image analysis of SEM records, X is the sliding distance, and  $h_s$  is the height of the scratch.

Special care was taken in scratching the surface of the diamond layer, where low loads 10 N-20 N were applied to avoid wrecking the tip of the indenter due to the high resistance of the material surface and a high degree of asperities.

Measured wear form microscratch test is compared to the predicted wear based on approaches in the literature by Rabinowicz (1996), Hutchings (1992), and Ning and